

SYNTHESIS AND CHARACTERIZATION OF LnFeO₃ PIGMENTS

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This work is focused on determination of the optimum firing temperature which leads to formation of the compounds LnFeO₃ with good pigment-application properties ($Ln=Gd, La, Yb, Tm, Lu$). Based on results of thermal analysis the compounds were prepared by the solid-state reaction at temperature 900 and 1000°C. Colours of pigments vary depending on type of used lanthanoid (from light sienna to dark brown) and also on the type of iron precursor. Generally, the compounds prepared from iron oxide have more interesting color properties. Colour of these compounds is brighter and deeper. Increasing of the calcination temperature from 900 to 1000°C causes the darkening of colour. The most probably, the darkening is connected with partial reduction of Fe³⁺ to Fe²⁺.

Keywords: heterometallic oxides, inorganic pigments, LnFeO₃, orthoferrites, thermal analysis

Introduction

The rare earth orthoferrites of the general formula LnFeO₃ (where Ln =lanthanoids) are one of the most useful materials for the applications such as catalysis [1], electrode materials and solid electrolytes for fuel cells and gas sensors [2, 3]. These compounds and metal ion substituted ferrites crystallized in perovskite orthorhombic structure in which the FeO₆ octahedrons are distorted and the extent of distortion depends on the ionic size of the rare earth [4].

The most conventional method for the preparation of these heterometallic oxides is the solid-state reaction of Ln₂O₃ and Fe₂O₃ single oxides. However, this method is very difficult to control the powder quality and particle size [2]. Usual ceramic method proposed by Pechini was used for preparation of sample LaFeO₃ which was studied from the pigment application point of view [5]. Other methods capable for synthesis of orthoferrites encompass thermal decomposition of heteronuclear complexes [2], sol–gel method [6, 7].

In this presented work the rare earth orthoferrites were prepared by solid state reaction [8–10] and the influence of synthesizing method on the pigment-application properties was studied.

Experimental

The compounds of general formula LnFeO₃ (where $Ln=Gd, La, Yb, Tm, Lu$) were prepared by solid-state reaction among Fe₂O₃ (TP-303; Precheza, a.s., CR) or FeOOH (Heubach, GmbH, Germany) and Ln₂O₃ (99% of purity; Indian Rare Earths Ltd., India). Raw materi-

als were mixing in a porcelain mortar in order to obtain homogenous reaction mixture which was calcinated at high temperature in an electric resistance furnace. The increase of temperature was set up on 10°C min⁻¹. The calcining temperatures 900 and 1000°C were maintained for 3 h. The pigments prepared were applied into organic matrix in mass tone and evaluated with respect to their colour properties by measurements of spectral reflectance in the visible region of light (400–700 nm) by equipment ColourQuest XE (HunterLab, USA). The measuring system uses colour space CIE $L^*a^*b^*$, where coordinate L^* expresses degree of brightness and coordinates a^* and b^* describe the colour hue. Thorough description of colour is done by recalculation of chrome extent C according to mathematical formula $C=(a^{*2}+b^{*2})^{1/2}$. The next measurement conditions were following: illuminant D65 (6500 K), 10° complementary observer and geometry of measurements $d/8^\circ$.

The particle size distribution of the samples was measured by equipment of Mastersizer 2000/MU (Malvern Instruments, UK). It is a highly integrated laser measuring system for analysis of particle size distribution. The equipment uses the scattering of incident light on particles. The solids were measured in solution of sodium hexamethaphosphate ($c=g\ l^{-1}$). The signal was evaluated on the basis of Fraunhofer bending.

Simultaneous TG/DTA measurements were performed by STA Jupiter 449 equipment (Netzsch, Germany) in the temperature range 30–1150°C at heating rate 10°C min⁻¹. The analysis was carried out in a ceramic crucible under air and α -Al₂O₃ was used as a reference material.

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The crystal structures of the powder materials were studied by X-ray diffraction analysis. The diffractograms of the samples were obtained by using the equipment diffractometer D8 (Bruker, GB) with a goniometer of 17 cm in the range 2θ of 10–80°. $\text{CuK}\alpha_1$ ($\lambda=0.15418$ nm) radiation was used for angular range of $2\theta < 35^\circ$ and $\text{CuK}\alpha_2$ ($\lambda=0.15405$ nm) for range of $2\theta > 35^\circ$. A scintillation detector was used.

Results and discussion

Thermal behaviour of reaction mixtures containing raw materials Ln_2O_3 and Fe_2O_3 or FeOOH were followed by methods of thermal analysis. The TG and DTA curves of the first reaction mixture (La_2O_3 and FeOOH) are shown in Fig. 1. The first mass loss (0.47%) is detected on the TG curve in temperature range 30–250°C. This process corresponds to the loss of remaining moisture and this effect is represented by endothermic effect on DTA curve with minimum at 123°C. The next two endothermic effects with minimum at 337 and 414°C detected on DTA curve are connected to the thermal dehydration of FeOOH [11, 12] and also with first step of thermal decomposition of lanthanum carbonate. Previous results of thermal analysis of lanthanum oxide proved that the compound is the mixture of lanthanum oxide and carbonate and chemical composition of the compound is characterised by formula $\text{La}_2\text{O}_3 \cdot 0.125\text{La}_2(\text{CO}_3)_3$ [13]. Thermal decomposition of lanthanum carbonate is three steps process, but the final step of the decomposition was detected only by TG curve with mass loss 0.59%. Formation of lanthanum orthoferrite is

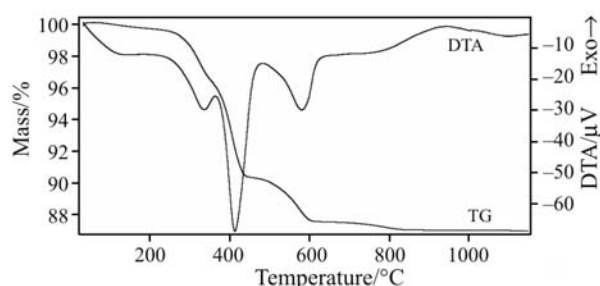


Fig. 1 DTA and TG curves of the decomposition of reaction mixture containing La_2O_3 and FeOOH (285.200 mg)

presented by exothermic peak with maximum at 943°C. The thermal decomposition of the second reaction mixture containing raw materials La_2O_3 and Fe_2O_3 is described in Table 1. Exchange of raw material of iron caused the increase of forming temperature of the new compound FeLaO_3 from 942 to 971°C. Thermal behaviour of reaction mixtures containing raw materials Gd_2O_3 and Fe_2O_3 or FeOOH is shown in Table 2. Several endothermic peaks were recorded on the DTA curve. These effects are connected to the partial decomposition of raw materials; especially, with dehydration of FeOOH and loss of oxygen and formation of lower gadolinium oxides $\text{Gd}_2\text{O}_{3-\delta}$. The endothermic effects detected at temperature higher than 900°C the most probably correspond to the partial reduction of Fe^{3+} ions to Fe^{2+} ions. The exothermic effects are connected with oxidation of gadolinium oxides $\text{Gd}_2\text{O}_{3-\delta}$ to $\text{GdO}_{2-\delta}$ and with beginning of formation of gadolinium orthoferrite. Thermoanalytical measurements show that formation of GdFeO_3 is under way at 820°C.

Table 1 Thermal decomposition of reaction mixtures $\text{La}_2\text{O}_3+\text{FeOOH}$ (Fig. 1) and $\text{La}_2\text{O}_3+\text{Fe}_2\text{O}_3$

$\text{La}_2\text{O}_3+\text{FeOOH}$			$\text{La}_2\text{O}_3+\text{Fe}_2\text{O}_3$		
$T_{\text{range}}/^\circ\text{C}$	Peak temperature/ $^\circ\text{C}$	Mass change/%	$T_{\text{range}}/^\circ\text{C}$	Peak temperature/ $^\circ\text{C}$	Mass change/%
30–250	123	–0.47	30–300	–	–0.14
250–350	337	–2.96	300–440	402	–5.87
350–450	414	–6.27	440–600	560	–2.87
450–600	581	–2.74	600–850	780	–0.57
600–850	–	–0.59	–	–	–

Table 2 Thermal decomposition of reaction mixtures $\text{Gd}_2\text{O}_3+\text{FeOOH}$ and $\text{Gd}_2\text{O}_3+\text{Fe}_2\text{O}_3$

$\text{Gd}_2\text{O}_3+\text{FeOOH}$			$\text{Gd}_2\text{O}_3+\text{Fe}_2\text{O}_3$		
$T_{\text{range}}/^\circ\text{C}$	Peak temperature/ $^\circ\text{C}$	Mass change/%	$T_{\text{range}}/^\circ\text{C}$	Peak temperature/ $^\circ\text{C}$	Mass change/%
30–200	145	–0.35	30–200	144	–0.31
200–400	335	–4.17	200–450	402	–1.10
400–800	410; 768	–0.91	450–800	768	–0.59
800–860	820; 849	–1.37	800–930	805; 925	–1.50
860–1000	923	–2.76	–	–	–

Table 3 Colour properties of the samples LnFeO₃ prepared by calcining at 1000°C

	Reaction mixture Ln ₂ O ₃ and FeOOH				Reaction mixture Ln ₂ O ₃ and Fe ₂ O ₃			
	<i>L</i> *	<i>a</i> *	<i>b</i> *	<i>C</i>	<i>L</i> *	<i>a</i> *	<i>b</i> *	<i>C</i>
LaFeO ₃	47.07	22.39	29.31	36.88	49.43	17.66	32.92	37.36
GdFeO ₃	50.22	21.85	31.51	38.34	52.23	18.77	35.75	40.38
YbFeO ₃	40.23	18.51	15.10	23.89	42.45	16.92	17.79	24.55
TmFeO ₃	41.72	19.38	17.87	26.36	41.73	16.81	17.89	24.55
LuFeO ₃	40.85	19.08	16.16	25.00	42.27	17.68	18.05	25.27

Colours of pigments vary depending on type of used lanthanoids and also on the type of iron precursor from light sienna and light brown to reddish brown (Table 3). Generally, the compounds prepared from iron oxide are brighter (*C*) and deeper (*L**), but on the other hand, pigments prepared from FeOOH contained bigger amount of red hue (*a**) and this characteristic makes them more interesting. On the basis of colour properties the pigments can be divided into two groups. In the first group of sienna pigments are compounds which contain lanthanide with bigger ionic radius; LaFeO₃ and GdFeO₃. Lanthanoids with smaller ionic radius, around 0.8 Å (Yb³⁺, Tm³⁺ and Lu³⁺), form pigments of reddish brown colour. Increasing of the calcination temperature from 900 to 1000°C causes the darkening of colour. The most probably, the darkening is connected with partial reduction of Fe³⁺ to Fe²⁺. The compound GdFeO₃, which was prepared by calcining of reaction mixture containing gadolinium oxide and iron hydroxide oxide at temperature 1000°C, provides the most interesting colour properties.

The calcining temperature also affects the particle size distribution of the pigments. While the *d*₅₀ values of the pigments prepared at 900°C are in range 2.0–3.5 μm, the pigments prepared at 1000°C have *d*₅₀ values in range 2.0–7.0 μm. From the pigment-application point of view is useful when the values of mean particle size (*d*₅₀) are less than 2 μm. This is connected with wider possibilities of practical usage and also smaller particles usually provide more interesting colour properties. This reason lead to the procedure of grinding. Compounds LaFeO₃ (1000°C) and GdFeO₃ (1000°C) were ground in planetary mill in presence of zirconium coruscules and ethanol medium for 10 and 20 min. Decreasing of particle size distribution of the compound LaFeO₃ prepared from reaction mixture containing FeOOH and La₂O₃ is clearly visible in Fig. 2. The mean particle size of the compound was reduced from value 3.42 to 1.66 μm after 20 min of grinding. This significant reduction of particle size distribution did not reflect on the changes in colour properties (Table 4). Gentle increasing of amount of yellow hue (*b**) and decreasing of red hue (*a**) caused the increasing of chrome (*C*).

Total colour difference between unground and 20 min ground samples was counted according to a formula $\Delta E_{CIE}^* = (\Delta L^{*2} + \Delta a^{*2} + \Delta b^{*2})^{1/2}$. Value $\Delta E_{CIE}^* = 1.3$ gives mathematical express of almost unperceptible colour difference. In the second case (reaction mixture Fe₂O₃ and La₂O₃), the best granulometric composition was obtained after 10 min of grinding (Fig. 3). The mean particle size was reduced from 2.34 to 1.42 μm and the grinding next 10 min was not effective. The colour properties of the ground samples were changed significantly. The value of coordinate *L** (brightness) decreased and it means the formation of darker colour hue. Also the coordinate expressing amount of yellow hue (*b**) decreased and it subsequently caused the formation of less rich colour. The value of total colour difference between unground and 10 min ground sample is 5.0. This value expresses the big colour difference which is perceptible by human eye.

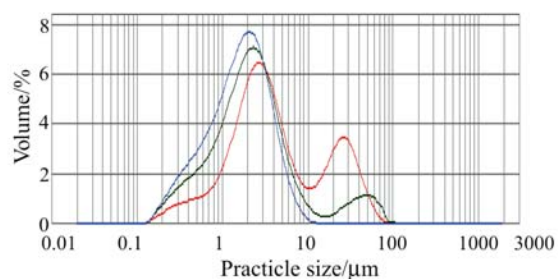
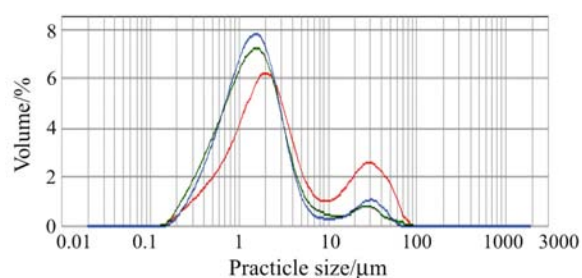
**Fig. 2** Particle size distribution of the samples LaFeO₃ (reaction mixture: La₂O₃+FeOOH); 1 – LaFeO₃; 2 – LaFeO₃ (10 min of grinding); 3 – LaFeO₃ (20 min of grinding)**Fig. 3** Particle size distribution of the samples LaFeO₃ (reaction mixture: La₂O₃+Fe₂O₃); 1 – LaFeO₃; 2 – LaFeO₃ (10 min of grinding); 3 – LaFeO₃ (20 min of grinding)

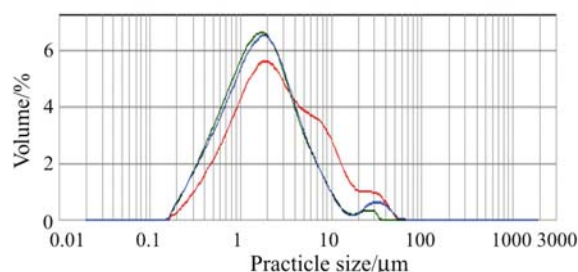
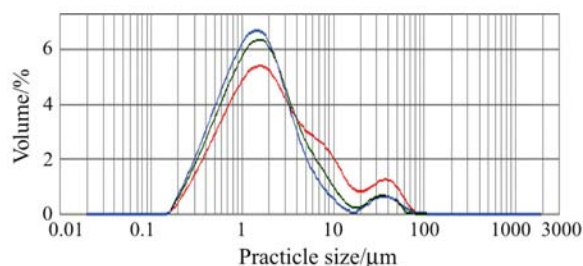
Table 4 Colour properties of the samples LaFeO₃ after grinding

Reaction mixture	Grinding time/min	<i>L</i> *	<i>a</i> *	<i>b</i> *	<i>C</i>	ΔE_{CIE}^*
La ₂ O ₃ +FeOOH	0	47.07	22.39	29.31	36.88	–
La ₂ O ₃ +FeOOH	10	46.63	22.25	28.11	35.85	1.29
La ₂ O ₃ +FeOOH	20	47.83	23.25	29.94	37.91	1.31
La ₂ O ₃ +Fe ₂ O ₃	0	49.43	17.66	32.92	37.36	–
La ₂ O ₃ +Fe ₂ O ₃	10	47.22	17.74	28.44	33.52	5.00
La ₂ O ₃ +Fe ₂ O ₃	20	46.65	17.37	28.05	32.99	5.62

Table 5 Colour properties of the samples GdFeO₃ after grinding

Reaction mixture	Grinding time/min	<i>L</i> *	<i>a</i> *	<i>b</i> *	<i>C</i>	ΔE_{CIE}^*
Gd ₂ O ₃ +FeOOH	0	50.22	21.85	31.51	38.34	–
Gd ₂ O ₃ +FeOOH	10	50.71	21.35	32.21	38.64	0.98
Gd ₂ O ₃ +FeOOH	20	50.54	21.34	32.17	38.60	0.90
Gd ₂ O ₃ +Fe ₂ O ₃	0	52.23	18.77	35.75	40.38	–
Gd ₂ O ₃ +Fe ₂ O ₃	10	51.01	17.40	32.67	37.01	3.59
Gd ₂ O ₃ +Fe ₂ O ₃	20	50.80	17.30	31.95	36.33	4.32

The similar results were obtained by grinding of GdFeO₃. The mean particle size of the product prepared from Gd₂O₃ and FeOOH was written down from 2.42 to 1.70 μm after 20 min of grinding (Fig. 4). The reduction of particle size distribution did not elicit strong change of colour properties (Table 5). Values of colour coordinates *L** and *a** almost did not

**Fig. 4** Particle size distribution of the samples GdFeO₃ (reaction mixture: Gd₂O₃ + FeOOH); 1 – GdFeO₃; 2 – GdFeO₃ (10 min of grinding); 3 – GdFeO₃ (20 min of grinding)**Fig. 5** Particle size distribution of the samples GdFeO₃ (reaction mixture: Gd₂O₃+Fe₂O₃); 1 – GdFeO₃; 2 – GdFeO₃ (10 min of grinding); 3 – GdFeO₃ (20 min of grinding)

shift and value of coordinate *b** (yellow hue) was slightly increased. It also caused the increasing of chrome (*C*). The grinding of the GdFeO₃, prepared from Gd₂O₃ and Fe₂O₃, has a negative effect on the colour properties. Values of all colour coordinates decreased with the decreasing of particle size distribution (Table 5). Grinding of the sample caused the change of mean particle size from 1.98 to 1.40 μm (Fig. 5) and formation the darker pigment with less amount of red (*a**) and yellow (*b**) hue and therefore with less chrome (*C*).

Structure of samples was studied by X-ray diffraction analysis. Samples prepared by calcining at 900°C are two-phases and contain next to the phase of rare earth orthopherrite unreacted rare earth oxide. Compounds LnFeO₃ crystallize in orthorhombic symmetry and values of lattice parameters of GdFeO₃ are *a*=5.349 Å; *b*=5.610 Å and *c*=7.667 Å. Values of lattice parameters of LaFeO₃ are *a*=5.553 Å; *b*=5.563 Å and *c*=7.862 Å.

Conclusions

The main aim of the research was to prepare and study pigmentary-application properties of the rare earth orthopherrites of general formula LnFeO₃. Adding Ln³⁺ ions to Fe₂O₃ change the colour of the pigment due to formation of perovskite compounds. Formation of new compounds with perovskite structure begins at temperature around 800°C in case of solid state reaction of oxide precursors. Ions of lanthanum and gadolinium form pigments of sienna or orange colour. Adding Lu³⁺, Tm³⁺ and Yb³⁺ to Fe₂O₃/FeOOH

forms pigments with dark reddish brown colour. The increase of temperature from 900 to 1000°C causes the creation of pigments with darker colour hue probably due to reduction of Fe³⁺ to Fe²⁺. Colours of pigments also vary depending on type of used iron precursor. Pigments prepared from iron oxide are brighter and deeper, but pigments prepared from FeOOH contain bigger amount of red hue and they are more appropriate for applications that require main particle size less than 2.0 µm.

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